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# Effect of Non-Uniform Temperature Exposure on the Out-of-Plane Bending Performance of Ordinary Laminated Glass Panels

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## **Effect of Non-Uniform Temperature Exposure on the Out-of-Plane Bending Performance of Ordinary Laminated Glass Panels**

#### **Abstract**

 Among the open design issues for structural laminated glass (LG) elements, their mechanical performance during fire exposure stands as a safety concern. This issue is the focus of this paper. An experimental investigation is conducted to evaluate the effect of non-uniform thermal gradients on the out-of-plane bending capacity of standard LG specimens. The pre-fracture, cracking, and post-fracture performances of LG panels were examined using a three-point bending setup. The bending tests were carried out on un- heated LG samples and heated samples, which were either hot or cooled down. The reliability of available analytical and numerical methods to predict the out-of-plane bending performance of LG is then investigated. The accuracy of the analytical and numerical methods was found to be highly affected by the non-uniform thermal boundaries, as well as by the assigned material properties, which are highly sensitive to the fire exposure.

 **Keywords**: Laminated glass (LG); out-of-plane bending; thermal exposure; thermal history; numerical modelling

#### **1. Introduction**

 The enhanced post-cracking performance of laminated glass (LG) has increased its use in architectural, automotive and defense applications in the last few decades [1]. Several research studies have focused on the in-plane and out-of-plane bending response of LG members under variable boundary and loading configurations. Those studies covered many issues including: (1) the definition of an effective thickness for multi-layered LG [2-4], (2) the development of novel treatment methods to improve the fatigue behaviour of LG [5-7], (3) the effects of interlayer type and stiffness on the flexural performance of LG [8-13], (4) the 8 effects of the interlayer materials and their bond characteristics [14, 15], (5) the delamination strength of LG and its effect on the structural performance [16, 17], (6) the performance of LG exposed to impact [18-20] and blast loads [21-23], and (7) the post-cracked structural behaviour [24-26]. The thermal breakage of glass during fire exposure has been experimentally investigated at the component and assembly levels [27-37]. Thermal fracture of glass is known to depend on various parameters, like frame

 properties, fire location, glass type, coating, etc. For clear glazing, it was established that the thermal load, panel size, smoke movement, and installation approach strongly affect the collapse mechanism and load- bearing capacity during fire. The effect of non-uniform fire exposure on clear and tinted monolithic glass panels was investigated by Vedrtnam et al. [38]. The analysis of local effects in the glass and interlayer components is the first step to conduct numerical thermal analyses [39, 40] and to calibrate simplified design procedures to examine glass thermal shock [41].

 The literature has mainly examined the thermal behaviour of vertical glass elements, which are used as curtain elements in buildings. The examined mechanical loads were, thus, minimum. Previous studies have mainly focused on thermal loading and did not address the major design challenges for glass elements exposed to mechanical loads. Bedon [42] summarized the current challenges, issues, and developments for glass systems exposed to fire, and provided a state of art on the material properties. The thermal conductivity and specific heat capacity of glass are known to increase with temperature, whereas the tensile strength and elastic modulus typically decrease [43]. This behaviour results in a progressive stiffness degradation, which complicates predicting the fire endurance for load-bearing glass elements [44]. The structural performance assessment of monolithic and LG requires detailed analysis of input parameters, including the thermo-physical and mechanical properties of materials, as well as thermal and mechanical boundaries.

 The effect of elevated non-uniform temperature exposure on the load-bearing capacity of LG members, as well as their breakage mechanism in both heated and post-heated conditions, is not largely addressed in the literature. Few literature studies related to horizontal glass members combined experimental and Finite Element (FE) numerical calculations to reproduce the complex thermo-mechanical phenomena [45, 46]. The non-uniform temperature distribution within LG panels results in specific thermal boundaries and mechanical stresses that should be carefully considered in their analysis and design. An accurate prediction  of the actual temperature gradient is a key parameter for design. Also, changes in material properties due to fire exposure need to be accounted for [42].

 The present research paper includes both original experiments on LG specimens and comparative calculations to study the above aspects. An experimental program is first discussed, in which a non-uniform thermal gradient is produced and then a traditional bending setup [38] is utilized to examine un-heated, heated, and heated & cooled LG specimens. A simplified analytical model [47] and more refined Finite Element (FE) numerical analysis utilizing ABAQUS [48] are then presented. The potentials and possible limits for both methods of analysis are discussed based on the experimental findings.

## **2. LG mechanical properties**

 Mechanical analysis of LG members depends on the composition of their layered sections [1, 2]. Given the viscoelastic characteristics of the interlayer, a multitude of research studies have been dedicated to the analysis of its mechanical properties and their effect on the overall structural capacity of the bonded glass panels. The flexural performance a typical LG panel, Figure 1a, is a function of the modulus of elasticity and shear modulus of the interlayer and the modulus of elasticity of the glass, which are affected by the temperature, as shown in Figures 1b and 1c. The shear modulus of the interlayer (*Gin*t), which is a function of the operational temperature and the strain rate, greatly affects the behaviour and results in fully rigid, semi rigid, or independent behaviour of the glass panels, as shown in Figure 1a. A high number of experimental, 19 analytical, and numerical investigations have been conducted to offer reliable material properties for PVB® and efficient calculation tools [49-52]. Having recognized that soda lime silica glass has a brittle elastic behaviour in tension and that the stiffness of bonding interlayer degrades with the increase of temperature, it 22 is accepted to design LG based on the assumption that  $G_{int} \rightarrow 0$  when the operational temperature is higher 23 than 40 °C, thus, the LG section is treated as two uncoupled glass panels. Limited experimental efforts are 24 available in the literature for LG components under temperatures higher than  $60^{\circ}$ C. The present research study addresses the shortage in the literature by experimentally examining the load-

bearing performance of LG elements, which were subjected to non-uniform thermal gradients. The analytical

and numerical studies account for thermo-mechanical modifications in the characteristics of interlayers and

the rigidity of the glass.



(a) Flexural stress distribution in LG section in case of full, null, and partial shear interaction



**Figure 1.** LG flexural behaviour [34].

# 

## **3. Experimental investigation**

## **3.1 Non-uniform thermal exposure**

 Forty LG specimens were tested. Each specimen consisted of a composite section made up from two 5 mm 6 thick annealed glass layers and a middle 0.76 mm thick PVB interlayer. The LG samples were manufactured by Safex\* Hindustan glass work limited, Prayagraj, India. The nominal length (*L)*, width (*B)*, and thickness (*h)* of the samples were 172 mm, 40 mm, and 10.76 mm, respectively. As per the 9 manufacturer, the glass composition was  $74.00\%$  SiO<sub>2</sub>, 13.00% Na<sub>2</sub>O, 10.50% CaO, 0.20% MgO, 1.30%

Al2O3, 0.30% K2O, 0.20% SO2, 0.40% Fe2O3 and 0.10% TiO2.

For the present study, a specialized test setup was constructed to allow exposing vertically installed LG

samples to realistic fire situations [38]. The test apparatus, shown in Figure 2, was prepared using a cast iron

- 13 burner and a set of aluminium plates, and can accommodate five LG samples simultaneously. The burner is
- 14 located at the middle of the setup almost at the same elevation as the bottom of the vertically installed glass
- **panels.** The full experimental program was carried out in a laboratory facility with negligible influence of
- airflow. The LG samples were placed between the aluminium plates, which had a gap (*t*gap) of 12 mm.

 The forty LG specimens were divided into 4 groups, and each group was exposed to different fire duration (*t*exp of 180, 360, 540, or up to 1200 seconds). The time-temperature relationships of the exposed (E) and unexposed (U) sides were recorded at the ten points, shown in Figure 3 (5 points per side), using J-type thermocouples and non-contact infrared thermometers.









Z



**Figure 3.** Schematic representation of reference control points for thermal measurements in LG specimens.

 Figure 4 presents the measured variation of temperature with time for the tested LG specimens. The maximum and minimum recorded temperatures on both the exposed (E) and unexposed (U) sides are shown. 10 The difference in temperature between the different samples considering the same point of measurement on 11 E or U sides was less than  $10^{\circ}$ C. Based on Figure 1 and from a mechanical point of view, the measured peak temperature, about 350°C, can be expected to correspond to a decrease of 5% in the modulus of elasticity *E* of glass and a full degradation of the shear rigidity of the PVB interlayer.





**Figure 4.** Time-temperature measurements for the tested LG specimens at different points on the exposed and unexposed surfaces.

Figures 5 and 6 show the variation of temperature along the height of the samples for the exposed and unexposed sides, while Figure 7 summarizes the overall gradient based on average measurements. The

distance *z* represents the distance from the base to the point of measurement, as shown in Figure 2. The maximum average temperatures of the E-side were 120 ºC, 205 ºC, 345 ºC and 360 ºC after *texp* of 180, 360, 540 and 1200 seconds, respectively. For the U-side, the maximum average temperature after 1200 seconds was 340 °C. The temperature difference in Figure 7 reached 25 °C near the top edge of the samples. The PVB layer was inspected after each fire test, and no visual damage was observed up to duration of 540 seconds. Post 540 seconds, the interlayer was found to start melting. Thus, the samples exposed to fire for 540 s has been chosen for the flexural tests. For duration of 1200 seconds, the PVB layer had completely melted.



**Figure 5.** Variation of the temperature of the exposed surface with the nondimensional height (*z*/*L*).



**Figure 6.** Variation of the temperature of the unexposed surface with the nondimensional height (*z*/*L*).



3 **Figure 7.** Measured temperature difference between the exposed and unexposed surfaces along the height of 4 the LG specimens.

#### **3.2 Flexural Tests**

- The experimental investigation followed the recommendations of ASTM D790-03 [38]. Flexural tests were performed on the fire-exposed LG samples as well as on five additional samples, which were not exposed to 4 fire. The ten samples that were exposed to fire for a duration of  $t_{exp}$ =540 seconds were flexurally tested to
- capture the effect of the heated interlayer. The experimental setup is shown in Figure 8. The simple supports
- were positioned at 5 mm from the ends of the total span *L*. The load *F* was applied in a displacement-control
- protocol with a rate of 1 mm/min. The corresponding vertical deformation was recorded at the mid-span
- section of LG samples. The laboratory condition at the time of experiments was characterized by an average
- 9 temperature of 28-31 °C.
- Following the thermal stage in Section 3.1, the flexural tests were separately conducted on un-heated, heated
- & cooled down samples, and hot samples. The hot condition was achieved by moving the heated LG samples
- 12 after fire exposure to the flexural machine in less than a minute. The heated  $\&$  cooled LG samples were
- flexurally tested after 30 minutes from the end of the fire exposure, which allowed them to cool down to the
- room temperature.
- 



**Figure 8.** Three-point bending performance assessment of LG specimens.

 The experimental force-displacement diagrams are shown in Figure 9, while the fracture patterns are shown in Figure 10. The duration needed to fracture the unheated glass samples was 1-3 minutes. For the heated samples, the duration reached 15 minutes due to the post-fracture residual ductility of the PVB.

22 The un-heated LG specimens cracked at a load of 280 to 574 N, which corresponded to a displacement of 1.2

- to 2.4 mm. The post-cracking performance was not consistent among the test specimens, as specimens #3
- and #4 failed at relatively lower loads. The fracture was characterized by multiple cracks distributed over a
- relatively wide fracture zone, Figure 10(a). Considering specimens heated & cooled (with *texp*= 540 seconds),
- their overall flexural performance was similar to the un-heated specimens. The load-bearing performance is
- shown in Figure 9(b). The fracture mechanism for heated & cooled specimens was found to affect a
- relatively smaller width of the total specimens, Figure 10(b). Specimens, which were tested while being hot,

 showed a considerable deviation in the measured load-displacement behaviour, Figure 9(c). This behaviour can be attributed to the PVB interlayer, which was partially weakened during the heated stage and could not return to its normal condition before the flexural test. This resulted in a relatively lower interlayer shear 4 transfer between the glass panels, and thus, an increased deformation for the tested specimens. Another 5 possible explanation could be that the melted interlayer tends to close the glass cracks and prevent 6 them from opening. Their peak load, 340 to 370 N, and the corresponding displacement, 1.4 to 1.6 mm, were similar to un-heated specimens. However, their fracture was reached at much higher displacement, which ranged between 11.5 to 11.8 mm. This performance can be useful in fire conditions as the glass panel can remain intact during fire for a longer period, which will prevent additional oxygen from supplying the fire, and thus, delay the flashover phenomenon.





(c) Specimens exposed to fire and tested in flexural while being hot **Figure 9.** Force-displacement curves from the bending setup.





(a) Specimens unexposed to fire (b) Specimens exposed to fire and then cooled prior to flexural testing



(c) Specimens exposed to fire and tested in flexural while being hot **Figure 10.** Fracture pattern of LG samples after flexural test.

 From Figure 9, the mechanical performance of LG samples was assessed in terms of average fracture load, deflection at fracture, and bending stiffness. The results for the un-heated and heated & cooled specimens, Figure 11, are very similar. This finding confirms the relatively null residual effect of thermal exposure on the mechanical properties of glass. The specimens, which were flexurally tested while being hot, showed lower variability in the values of the failure load and much higher displacements at failure. The lower variability might be attributed to limited influence of microcracks/impurities of the glass and dominance of the properties of the softened interlayer in determining the flexural behavior. However, it is worth mentioning that average maximum load before fracture was lower in the heated condition, as the interlayer

 was significantly affected at the boundaries of specimens, which played a significant role in determining the flexural strength. For the same reason, the higher displacement at failure for the heated condition was achieved because of the softened interlayer, which kept the fractured glass intact for longer duration.







**Figure 11.** Force and displacement for the LG specimens at collapse.

 From the collected force-displacement records, the bending stiffness *EJ* of composite LG samples was also calculated. Experimental *EJ* values for the tested samples are summarized in Figure 12. The specimens, which were heated and then cooled prior to the flexural test, had a slight reduction in their stiffness of about 0.95%. The specimens, which were hot during the flexural test, showed a higher reduction of about 1.98%. Figure 12 clearly shows some variation between the *EJ* values within each group of samples, which reflects the variability of the value of *Gint* and *E* for glass under different thermal scenarios. It is also possible to notice that fire did not cause marked modification in the average bending stiffness *EJ*. The major effect was on the weak bond offered by PVB. This behaviour is further explained in Section 4. Figure 12(d) shows in fact the overall variation of *EJ* with the imposed thermal condition. It can be perceived from average and min/max *EJ* estimates that specimens flexurally tested in hot conditions offered

 the lowest stiffness variability. The scatter of measured *EJ* values progressively increases for samples bended in cold conditions and suggests some further influence of *EJ* due to the viscous response of the interlayers at room temperature. For the heated & cooled samples, Figure 12(d), the average prediction is still in line with the un-heated samples, but again the max/min interval is minimized.



1 **Figure 12.** Analytical flexural stiffness of LG specimens.

## 3 **4. Analytical Investigation**

## 4 **4.1 Modelling**

5 The application of the equivalent thickness (EET) approach to estimate the flexural stiffness is investigated 6 in this section. For double LG panels, the equivalent thickness  $(h_{ef,w})$  is estimated using Eq. (1) [39].

7 
$$
h_{ef,w} = \sqrt[3]{\frac{1}{\frac{\eta}{h_1^3 + h_2^3 + 12I_s} + \frac{1 - \eta}{h_1^3 + h_2^3}}}
$$
(1)

8 where  $h_1$  and  $h_2$  are the thicknesses of glass layers (5 mm),  $h_{int} = 0.76$  mm is the thickness of interlayer,  $I_s$  is 9 the composite action effect evaluated using Eq. (2), and  $\eta$  denotes the shear bonding level and is given by 10 Eq. (3).  $\eta$  ranges from 0 for weak connection to 1 for fully rigid connection.

$$
1 \tI_s = \frac{h_1 h_2}{h_1 + h_2} \cdot [h_{int} + 0.5(h_1 + h_2)]^2 \t(2)
$$

$$
2 \qquad \eta = \frac{1}{1 + \frac{E h_{int} J_1 + J_2}{G_{int} B} J_{tot} A_1 + A_2} \tag{3}
$$

In the above equations,  $E$  is the modulus of elasticity of glass,  $G_{int}$  is the secant shear modulus of the PVB 5 interlayer (for the specific temperature and loading condition),  $B = 40$  mm is the width of the panel, 6  $\Psi$  is a coefficient that accounts for the boundary conditions and load distribution [39] = 10/ $L_0^2$  (with *Z*  $L_0 = L - 5$  mm ×2 = 162 mm, the actual bending span),  $J_i$ ,  $A_i$ , and  $J_{tot}$  are given respectively by:

$$
8 \t J_i = \frac{Bh_i^3}{12} \t (i = 1, 2) \t (4)
$$

$$
9 \qquad A_i = Bh_i \qquad (i = 1, 2) \tag{5}
$$

10 
$$
J_{tot} = J_1 + J_2 + \frac{A_1 A_2}{A_1 + A_2} \cdot [h_{int} + 0.5(h_1 + h_2)]^2
$$
 (6)

11

 The obtained equivalent thickness variation is shown in Figure 13 as a function of *Gint*. It can be seen that for an average room temperature of 28-30°C, the calculated *hef,w* is very close to the lower "Abs" limit condition, due to the presence of a relatively weak bond (*Gint*≈ 4.5MPa the expected shear modulus based on Figure 1). 15



16

17 **Figure 13.** EET variation as a function of *Gint*.

18

#### 19 **4.2 Results and experimental comparison**

20 For the analytical calculations carried out on the unheated samples, annealed glass was assumed to be 21 temperature-independent and having a linear elastic stress-strain curve with *E*= 70 GPa. The initial 22 composite bending stiffness *EJ* for the tested specimens was calculated analytically and compared to the

 experimental results. The mid-span displacement *fmax* due to the imposed vertical force *F* was calculated using Eq. (7) and its variation is shown in Figure 14.

$$
3 \qquad f_{max} = \frac{FL_0^3}{48EI} \tag{7}
$$

4 where  $E J = f(h_{e f w})$ .

 For the un-heated LG samples, the interlayer was first assumed to provide a weak shear connection, with *Gint* of zero ("Abs" case). To ensure accurate analytical solutions, it is needed to account for the loading scenario, temperature, humidity, thickness of constituent layers, and the PVB properties. These factors cannot be realistically estimated for different LG samples; thus, it seems adequate to use a weak bond condition for the interlayer and nominal thicknesses for all the LG components. For comparisons, analytical calculations are also repeated with a rigid shear connection in which *Gint* approaches the shear rigidity of glass, "Full" case in 12 Figure 14, and an intermediate case, with  $G_{in} = 4.5$  MPa, was also considered. The resulting analytical flexural responses, shown in Figure 14, and their comparison to the five tested

 samples clearly indicate that the weak bonding condition provides a good estimate for the initial stiffness of tested LG specimens. Minor deviation can be seen for cases of *Gint*= 4.5 MPa and zero, which can be due to the modified interlayer properties, as well as the minor deviation of the real thickness from the nominal values. Regardless, the analytical model can correctly reproduce the elastic flexural response of un-heated 18 LG samples at ambient temperatures. It should be noted that the model suggested in [55] can be modified to

- 19 analyze bending behaviour of cracked LGs.
- 



**Figure 14.** Flexural response of un-heated LG specimens.

 The same analytical model and procedure was then examined for LG samples subjected to thermal cycles. Following the preliminary un-heated calculations in Figure 14, a weak "Abs" shear connection was again assumed to represent the PVB foils. For this calculation step, the LG panel is assumed to have to a uniform temperature. This approach represents a lower limit condition for the examined LG samples, given that the temperature evolution is typically non-uniform (as in Section 3), and thus the corresponding degradation of material properties (i.e. Figure 1) is also non-uniform. The effect of temperature on the glass panel was taken into consideration using the experimental *E* values, given in Figure 1. For an imposed uniform temperature, 8 the modified *E* value is estimated from Figure 1, and then  $h_{eq,w}$  is estimated using Eq. (1). Because of the 9 variation of the temperature with time, the material properties change, and calculations based on Eq. (1) need to be repeated. In this manner, it is possible to predict the bending stiffness *EJ* of the heated LG sample, and its modification with thermal boundaries. The resulting *EJ* values are shown using the blue curve in Figure 15. Due to the assumptions of uniform temperature within the LG section, a significant reduction in the *EJ* 13 value can be observed at about 100 °C (-12%). For higher imposed temperatures, the *EJ* degradation is still linear but less pronounced than in the first 100°C. A -18% reduction is predicted at a temperature of 400°C. For comparisons, the experimental (average) *EJ* values are also presented in Figure 15. It is clear that the adapted EET analytical model is able to predict the *EJ* values for the un-heated and the cooled-down samples, with reasonable accuracy. This is due to the negligible thermal effects of the relatively high but short-term fire exposure (*texp*= 540 s). However, for the heated samples, the predicted *EJ* value in Figure 15 is 10% lower than the corresponding experimental results. This lower estimate can be attributed to the uniform temperature assumption in LG, or the used *E* values for the glass [34, 35].





23 **Figure 15.** Expected analytical flexural stiffness variation with temperature for the tested LG specimens.

#### 1 **5. Numerical Investigation**

#### 2 **5.1. Modelling of heated specimens**

3 The tested specimens were modeled using ABAQUS software like in a previous research step of the project 4 [30], in which a similar test setup was used for monolithic samples in place of LG specimens. The analysis 5 was composed of a thermal stage (s1) to impose the **experimental** temperature scenario (heating stage), 6 followed by a coupled static stage  $(s2)$  to calculate the thermal stresses in the burner setup due to temperature 7 variation, and another mechanical loading stage (s3) to examine the flexural behaviour. The adopted general 8 procedure of simulation is summarized in Figure 16. The thermal variation effect of metal slots in stages s1 9 and s2 was disregarded, based on earlier studies [30]. Besides, variations with temperature in the mechanical 10 properties of the different materials were considered.





12 **Figure 16**. (a) Schematic procedure for thermal and mechanical FE numerical analysis of LG samples under 13 different thermal cycles and (b) mesh detail.



1 modeled in their nominal geometry using DC3D8 brick elements for the full thermal analysis (s1 to s3 steps).

2 A regular mesh pattern with 2 mm average edge size and 4 solid elements in the thickness of each glass

3 layers (2 in the thickness of PVB) was used. A total of 16,236 bricks for LG and 7,026 bricks for the metal

4 parts (103,842 DOFs in total) were used. The final layout is shown in Figure 16(b).

 A rigid temperature-independent connection was utilized at the interface of the glass and PVB layers. This 6 means that possible delamination effects were disregarded, but material degradation with temperature was 7 only considered. The thermo-physical and mechanical characterization of glass are shown in Figure 17, which were utilized by Kozłowski and Bedon [35]. The moduli of glass and PVB were assumed to follow the modifications presented in Figures 1(b) and 1(c). Temperature-independent material properties were











2 were taken into account over an exposure time of *texp*= 540 seconds. To reproduce the real non-uniform 3 thermal boundaries. These measured temperatures were imposed to both the E and U surfaces of glass. After 4 the fire exposure in  $s1$ , the FE assembly was moved to the mechanical stage to capture thermal stresses in the 5 components. The bending loading stage was then carried out for the simply supported LG samples, either in 6 hot conditions (s2) or at room temperature after cooling (s3). Cooling was assumed to take place in the same 7 simply supported bending setup. 8 In both cases, the uncoupled mechanical analysis was carried out replacing the original brick components of 9 glass and PVB with C3D8R elements, having identical mesh pattern as in the thermal model (Figure 16b). 10 The temperature-independent metal parts were only considered in the mechanical stage s1 to account for the 11 potential thermal stresses at the edges of glass. The metal parts were then physically deleted from the FE 12 assembly and replaced by nodal boundaries to reproduce the setup shown in Figure 8. In such a procedure, 13 the temperature nodal records and evolutions in glass and PVB from stage s1 (and / or s2) were used in the 14 bending stage (s2 or s3) to account for the non-uniform thermal exposure of heated or heated  $\&$  cooled 15 specimens, respectively. 16 In doing so, for both stages s2 and s3, no experimental temperatures were imposed to the heated models after 17 stage s1. Cooling / bending was thermally simulated assuming an average laboratory temperature of 28  $\degree$ C.

1 For the heating thermal stage (s1), the input experimental temperatures of the 10 controls points in Figure 3

18 The heat transfer boundary conditions for radiation and convection were, thus, defined as incident radiation 19 from a black body [35]. Following [53, 54], the heat transfer coefficient was set to 25 W/m<sup>2</sup>K for the glass 20 surface and  $8.02 \text{ W/m}^2\text{K}$  for the unexposed surfaces. A standard emissivity coefficient equal to 0.95 was 21 used [53].

22

#### 23 **5.2. Results**

24 Following the previous experimental and analytical discussions, major attention is spent in the present 25 section for specimens bended under hot conditions  $(s2)$ . Given that the heated LG specimens were tested 26 flexurally immediately after the fire exposure  $(s1)$ , no intermediate cooling effects were considered for FE 27 modelling.

28 The FE results in Figure 18 represent the time instant at which the LG samples are removed from the burner.

29 Figure 18(a) shows the contour plot of imposed temperatures for the unexposed side after t<sub>exp</sub> of 540 seconds,

30 while Figure 18(b) summarizes the typical temperature evolution at the 10 control points of the FE assembly

31 for the E and U sides of the glass panel. The simplified FE procedure is clearly able to match the

- 32 experimental scenario. The temperature and thermal stresses for PVB are shown in Figures 18(c) and (d).
- 33 The PVB temperature is found to be of the same order of magnitude as the E and U sides of glass panels.
- 34 The peal local stresses, observed in Fig. 18(d), are similar to the experimental observations, shown in Figure

1 10(c). The top glass panel experienced the highest temperatures, and thus major stress concentrations, as 2 shown in Figure 18(e). The maximum principal stresses were numerically measured as 19 MPa.

3 For the mechanical loading stage, the boundary conditions matched those of the experimental setup. The

4 effect of the fire exposure was reflected in the input temperature-dependent mechanical properties of glass 5 and interlayer materials, as well as by the residual thermal stresses in Figure 18.

6 Figure 19 shows the typical bending response. While point A marks the area that experienced the highest 7 temperature in the heating stage, point B corresponds to the maximum tensile stress during the mechanical loading stage. The mechanical loading stage, see Figure 19(a), followed the *texp* phase and included the ambient cooling of materials while loading. The force-displacement response is shown in Figure 19(b), which roughly captures the experimental observations but still to lack the local effects happening due to the changes in the PVB properties. Figure 19(c) shows that the fracture initiation is expected to initiate at the mid-span region of glass panels. The inverse analysis of fracture stresses at the edge suggests a material strength in the order of 20 MPa. Given the presence of a mostly melted PVB interlayer, the same order of 14 magnitude of maximum stresses can be perceived for both glass panels, as shown in Figure 19(d). Such a stress magnitude is significantly lower than the nominal tensile bending strength of annealed glass (45 MPa 16 the characteristic value in cold conditions). At the same time, the estimated stress is in good correlation with thermal shock resistance values of literature for glass specimens. The average experimental thermal failure is 18 in the order of 15 MPa as reported in [42].

- 19
- 20





(a) Temperature in glass  $(a)$  Temperature at selected control points  $(U \text{ and } E \text{ sides})$ 



2 3





(d) Contour plot of maximum principal stresses at Force of 500N (bottom view) 1 **Figure 19.** Bending performance of the heated LG specimens (legend values in Pa).

#### **6. Conclusions**

 This manuscript reports out-of-plane bending behaviour of laminated glass (LG) members exposed to fire, based on experimental, analytical, and numerical approaches. Attention was focused on the different thermal 4 boundaries (non-uniform thermal exposure) and thermal **histories**, and their effects on the actual mechanical performance of LG element in out-of-plane bending. The conclusions of this study can be summarized in the following points.

- The out-of-plane bending performance of LG members was not affected by exposure to temperatures up to 400 °C, when they were mechanically tested after being cooled down.
- The small variation in the pre-cracked flexural stiffness of the tested specimens under different 10 thermal **histories** confirms the stability of thermo-physical and mechanical properties.
- During flexural tests, fracture was found to initiate in the lower glass plate near the mid-span section (load application), and then to move to the upper glass plate.
- The un-heated LG samples generally experienced multiple fractures during the flexural test and were thus characterized by a wider cracked zone at collapse.
- 15 The bending collapse for both the un-heated and heated & cooled LG specimens was mostly characterized by an abrupt failure mechanism, without any kind of residual stiffness nor ductility.
- 17 The heated LG samples which were tested flexurally in hot conditions proved to offer residual capacity in out-of-plane bending with deformations reaching ten times the other specimens. This response is due to the melted interlayer, which allowed the glass panels to stay intact during the flexural test. The pre-cracked bending stiffness of the LG specimens, furthermore, does not suggest 21 relevant modifications in the modulus of elasticity of glass.
- Both the analytical and numerical analyses confirmed the relevant effects of non-uniform temperature conditions on the LG specimens.
- 24 In the case of un-heated and heated & cooled specimens, the calculations were found to mostly agree with the load-deflection and bending stiffness experimental data, and to confirm the presence of a rather weak bonding connection between the glass layers.
- For the heated LG samples bended in hot conditions, major limits were observed in the analytical and numerical models to capture the progressive modification of material properties with temperature modification.
- 
- 
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